

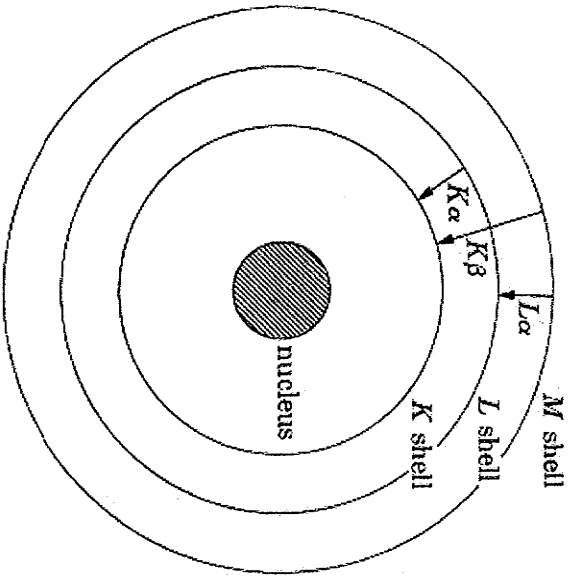
## **Characterization: X-ray**

# Scattering Techniques

incoming	outgoing	techniques
electron	electron	Auger electron spectroscopy (AES) electron diffraction low energy electron diffraction (LEED) reflection high-energy electron diffraction (RHEED) Scanning electron microscopy (SEM)
ion	ion	Transmission electron microscopy (TEM) Rutherford backscattering (RBS) Channeling
ion	target ion	secondary ion mass spectroscopy (SIMS)
x-ray	x-ray	x-ray fluorescence spectroscopy (XFS) x-ray diffraction
neutron	neutron	neutron diffraction
x-ray	electron	x-ray photoelectron spectroscopy (XPS)
UV	electron	UV photoelectron spectroscopy (UPS)
electron	x-ray	electron microprobe (EM) x-ray microanalysis
photon	electron	photoemission
electron	photon	inverse photoemission



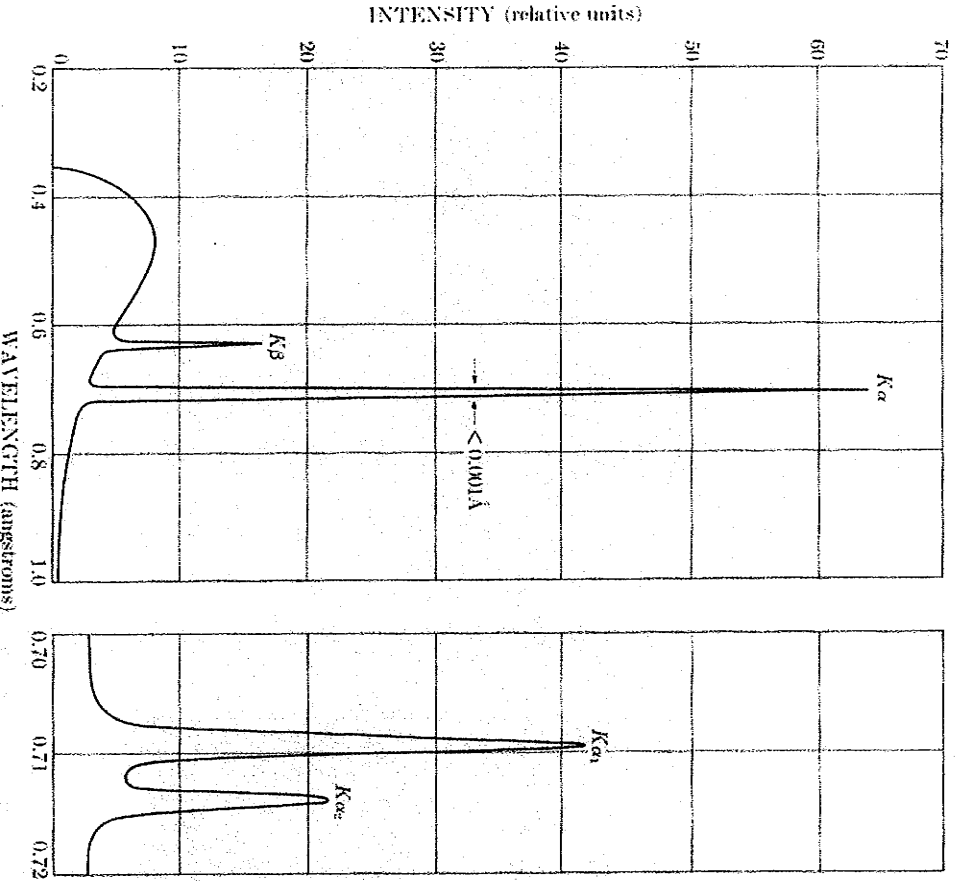
# Line Spectrum



Characteristic X-ray:

unique to element

incident  $e^-$  knocks  $e^-$  out of orbit of inner shell, outer shell  $e^-$  decays to fill

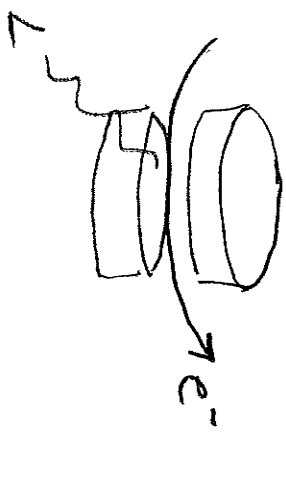


Spectrum of Mo at 35 kV (schematic). Line widths not to scale. Resolved  $K\alpha$  doublet is shown on an expanded wavelength scale at right.

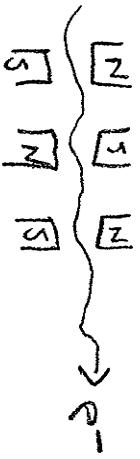
# Cu Lines

Hummer's

Synchrotron radiation  
(like Bremsstrahlung)

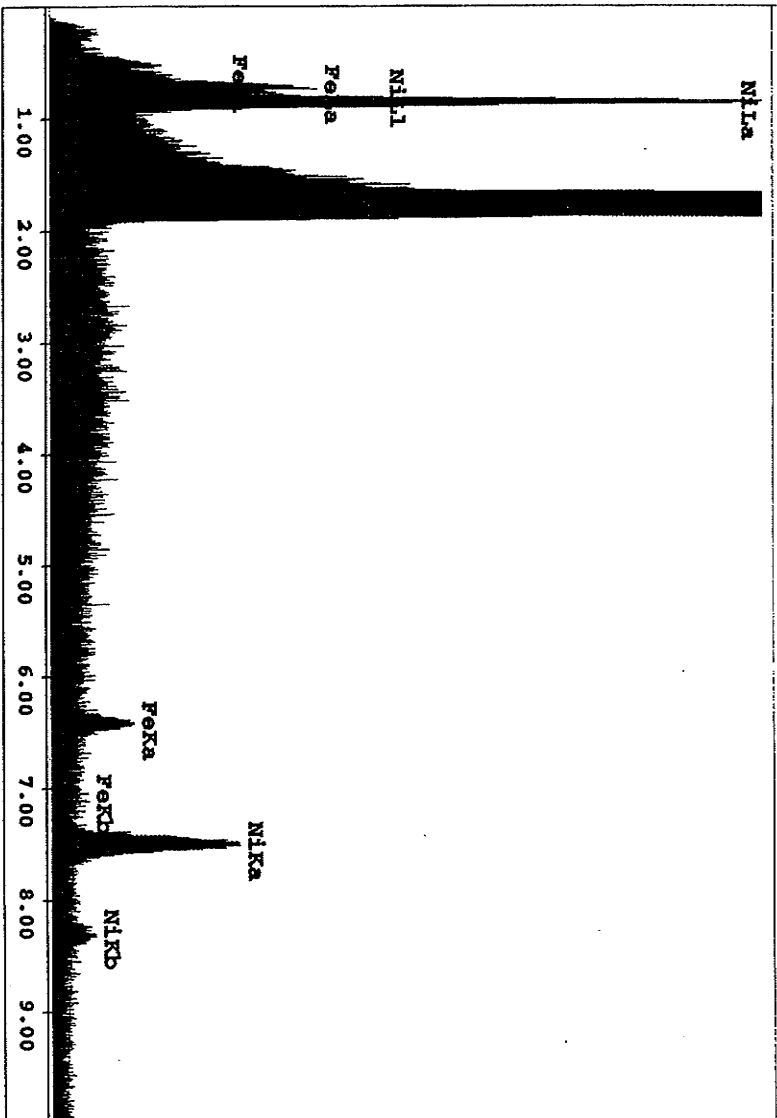


undulator or wiggler



	Siebahn	Designation	Energy in keV	Absorption Edge in keV		
K X-Rays	K $\beta$	K $\beta_1$	8.9770	8.98029 (K)		
		K $\beta_2$	8.90529			
		K $\beta_3$	8.9029			
	K $\alpha$	K $\alpha_1$	8.04778			
		K $\alpha_2$	8.02783			
		K $\alpha_3$	1.0228			
	L X-Rays	L $\beta$	L $\beta_1$		0.9498	0.93306(L $\beta_1$ )
			L $\beta_2$		0.9297	
			L $\beta_3$		0.832	
		L $\alpha$	L $\alpha_1$		0.8111	
L $\alpha_2$						
L $\alpha_3$						

# EDX



**EDX Peak Quantification (Standardless)**  
**Element Normalized**  
**SEC Table : Default**

Element	Wt %	At %	K-Ratio	Z	A	F
FeK	15.47	16.14	0.1813	0.9876	0.9885	1.2000
NiK	84.53	83.86	0.8192	1.0022	0.9671	1.0000
Total	100.00	100.00				

Element	Net Inte.	Bkgd Inte.	Inte. Error	P/B
FeK	2.50	1.55	5.47	1.61
NiK	8.07	1.39	2.36	5.82

# Absorption/Filtering

$$I_x = I_0 e^{-(\mu/\rho) \rho x}$$

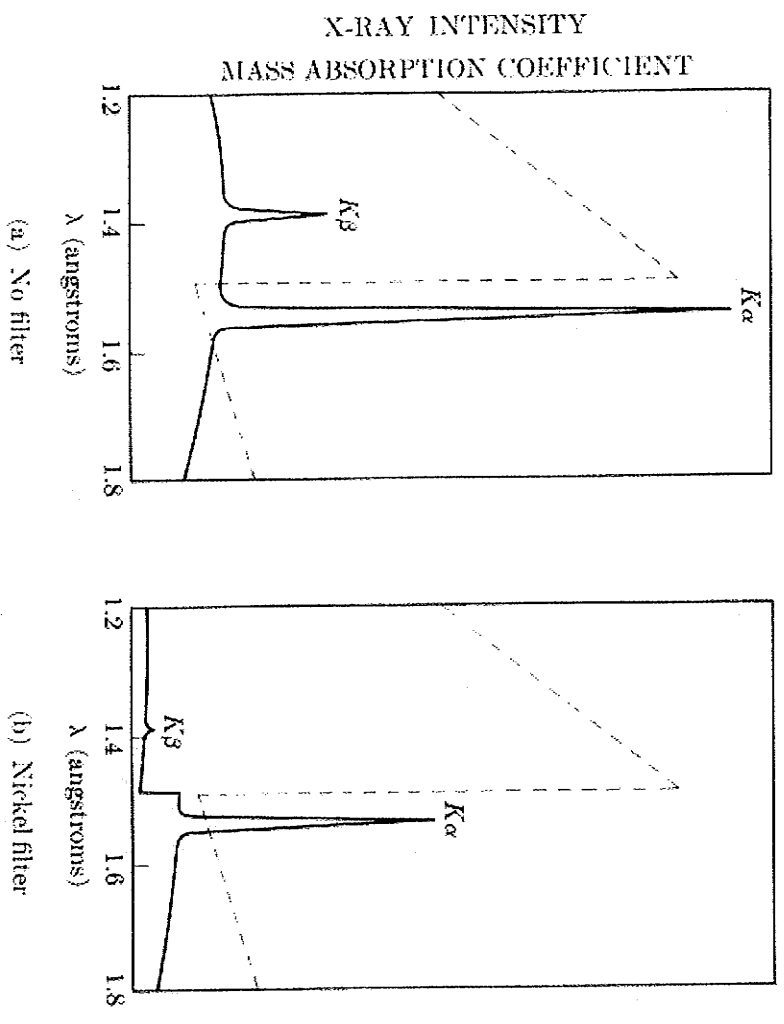
$\mu$  - Absorption coefficient (cm<sup>-1</sup>)

$1/\mu$  - One absorption length

$$\mu / \rho = k \lambda^3 Z^3$$

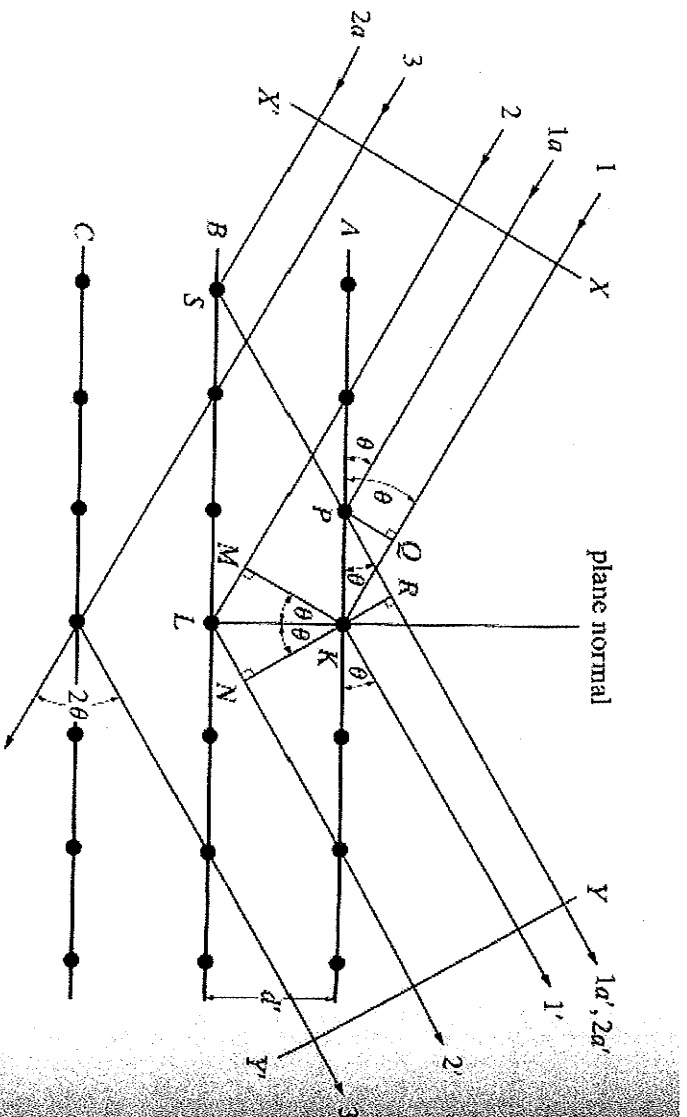
$x$ : thickness

Mass absorption coefficient



Comparison of the spectra of copper radiation (a) before and (b) after passage through a nickel filter (schematic). The dashed line is the mass absorption coefficient of nickel.

# Diffraction



Scalar description:

$$\text{Bragg's Law: } 2d \sin \theta = n\lambda$$

Vector description:

$$\text{Laue conditions: } \Delta \mathbf{k} = \mathbf{k}_o - \mathbf{k} = \mathbf{G} \text{ - necessary}$$

$$\text{q-value: } |\Delta \mathbf{k}| = 4\pi \sin \theta / \lambda$$



# d-Spacing

Cubic:

$$\frac{1}{d^2} = \frac{h^2 + k^2 + l^2}{a^2}$$

Tetragonal:

$$\frac{1}{d^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2}$$

Hexagonal:

$$\frac{1}{d^2} = \frac{4}{3} \left( \frac{h^2 + hk + k^2}{a^2} \right) + \frac{l^2}{c^2}$$

Rhombohedral:

$$\frac{1}{d^2} = \frac{(h^2 + k^2 + l^2)\sin^2 \alpha + 2(hk + kl + hl)\cos^2 \alpha - \cos \alpha}{a^2(1 - 3\cos^2 \alpha + 2\cos^3 \alpha)}$$

Orthorhombic:

$$\frac{1}{d^2} = \frac{h^2}{a^2} + \frac{k^2}{b^2} + \frac{l^2}{c^2}$$

Monoclinic:

$$\frac{1}{d^2} = \frac{1}{\sin^2 \beta} \left( \frac{h^2}{a^2} + \frac{k^2 \sin^2 \beta}{b^2} + \frac{l^2}{c^2} - \frac{2hl \cos \beta}{ac} \right)$$

Triclinic:

$$\frac{1}{d^2} = \frac{1}{V^2} (S_{11}h^2 + S_{22}k^2 + S_{33}l^2 + 2S_{12}hk + 2S_{23}kl + 2S_{13}hl)$$

In the equation for triclinic crystals,

$V$  = volume of unit cell (see below),

$$S_{11} = b^2c^2\sin^2 \alpha,$$

$$S_{22} = a^2c^2\sin^2 \beta,$$

$$S_{33} = a^2b^2\sin^2 \gamma,$$

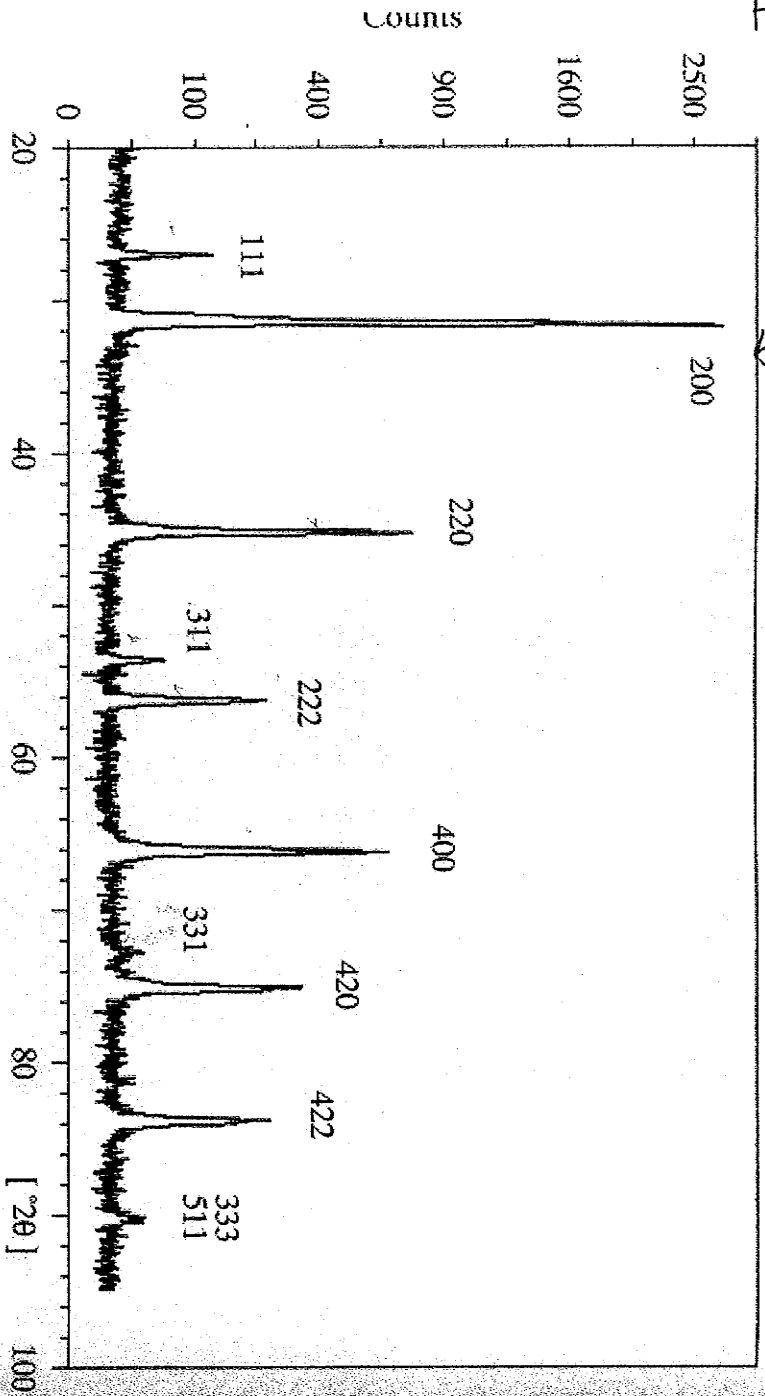
$$S_{12} = abc^2(\cos \alpha \cos \beta - \cos \gamma),$$

$$S_{23} = a^2bc(\cos \beta \cos \gamma - \cos \alpha),$$

$$S_{13} = ab^2c(\cos \gamma \cos \alpha - \cos \beta).$$

# Polycrystalline NaCl

Miller indices  
 give small  $h$  &  $k$   
 the planes, and  
 $h$  is  $\Delta d$  of  
 + vary  
 $\therefore$  small  $\theta$   
 associated with  
 lower Miller  
 indices.



Why no (210), (211)...?

# Structure Factor

$$F_{hkl} = \sum_1^N f_n e^{2\pi i(hu_n + kv_n + lw_n)}$$

$|F| = \frac{\text{amplitude of the wave scattered by all the atoms in a unit cell}}{\text{amplitude of the wave scattered by one electron}}$

e.g.: fcc lattice, 4 same atoms at (0, 0, 0), ( $\frac{1}{2}$ ,  $\frac{1}{2}$ , 0), ( $\frac{1}{2}$ , 0,  $\frac{1}{2}$ ) and (0,  $\frac{1}{2}$ ,  $\frac{1}{2}$ ).

$$F_{hkl} = f [1 + e^{\pi i(h+k)} + e^{\pi i(k+l)} + e^{\pi i(l+k)}] \neq 0 \text{ if } h, k, l \text{ are all odd or all even}$$

Allowed diffraction:

fcc: all odd or all even

bcc:  $h+k+l = \text{even}$

# Allowed Diffractions

$h^2 + k^2 + l^2$	Cubic				Hexagonal	
	hkl		Diamond	hk	$h^2 + hk + k^2$	hk
	Simple	Face-centered				
1	100	...	110	...	1	10
2	110	...	...	...	2	2
3	111	111	...	111	3	11
4	200	200	200	...	4	20
5	210	...	...	...	5	...
6	211	...	211	...	6	...
7	...	...	...	...	7	...
8	220	220	220	220	8	21
9	300, 221	...	...	...	9	30
10	310	...	310	...	10	...
11	311	311	...	311	11	...
12	222	222	222	...	12	22
13	320	...	...	...	13	31
14	321	...	321	...	14	...
15	...	...	...	...	15	...
16	400	400	400	400	16	40
17	410, 322	...	...	...	17	...
18	411, 330	...	411, 330	...	18	...
19	331	331	...	331	19	32
20	420	420	420	...	20	...
21	421	...	...	...	21	41
22	332	...	332	...	22	...
23	...	...	...	...	23	...
24	422	422	422	422	24	...
25	500, 430	...	...	...	25	50
26	510, 431	...	510, 431	...	26	...
27	511, 333	511, 333	...	511, 333	27	33
28	...	...	...	...	28	42
29	520, 432	...	...	...	29	...
30	521	...	521	...	30	...
31	...	...	...	...	31	51
32	440	440	440	440	32	...
33	522, 441	...	...	...	33	...
34	530, 433	...	530, 433	...	34	...
35	531	...	531	531	35	...
36	600, 442	600, 442	600, 442	...	36	60
37	610	...	...	...	37	43
38	611, 532	...	611, 532	...	38	...
39	...	...	...	...	39	52
40	620	620	620	620	40	...
41	621, 540, 443	...	...	...	41	...
42	541	...	541	...	42	...
43	533	533	533	533	43	61
44	622	622	622	...	44	...
45	630, 542	...	...	...	45	...
46	631	...	631	...	46	...
47	...	...	...	...	47	...
48	444	444	444	444	48	44
49	700, 632	...	...	...	49	70, 53

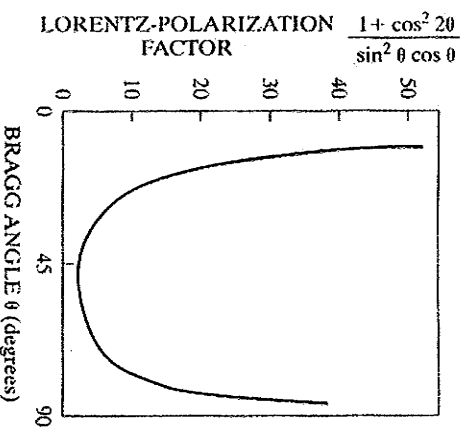
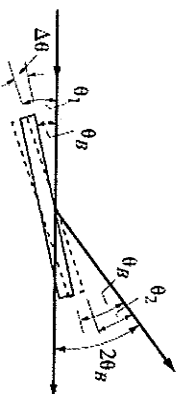
# Diffraction Intensity

$$|F|^2 p \left( \frac{1 + \cos^2 2\theta}{\sin^2 \theta \cos \theta} \right) A(\theta) e^{-2M}$$

Structure factor F

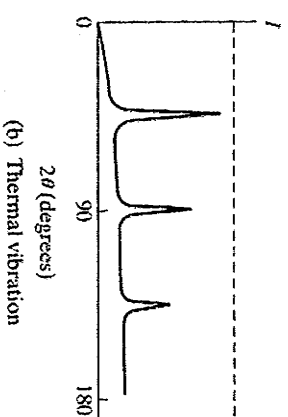
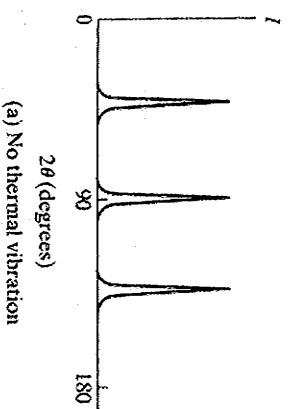
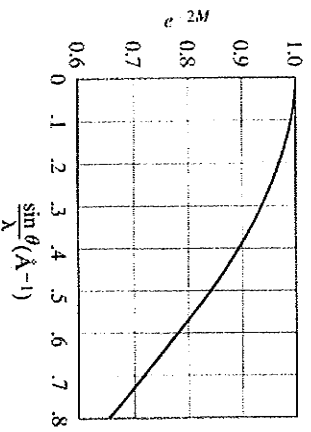
Multiplicity factor p

Lorentz factor



Absorption factor A(θ)

Debye-Waller (temperature) factor



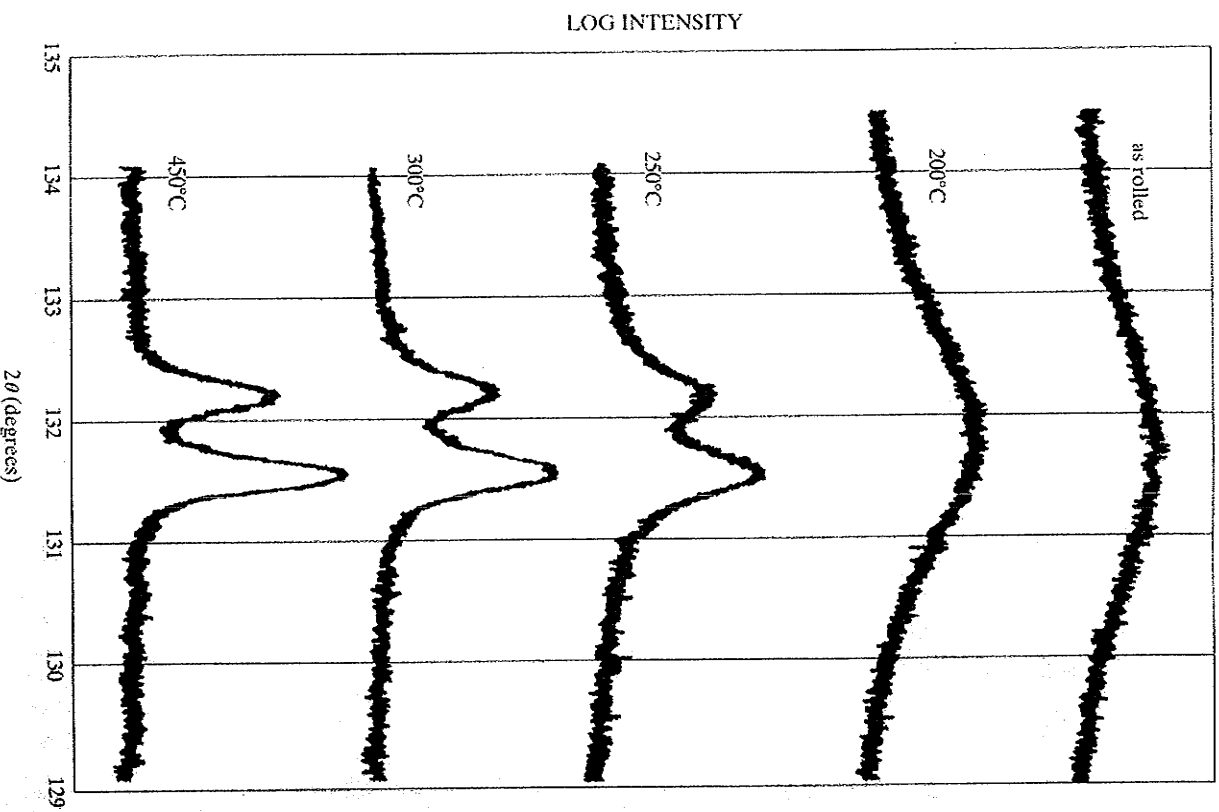
# Multiplicity Factor

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Cubic:	$hkl$	$hhl$	$0kl$	$0kk$	$hkh$	$00l$			
	$\frac{48^*}{1}$	$\frac{24}{1}$	$\frac{24^*}{1}$	$\frac{12}{1}$	$\frac{8}{1}$	$\frac{6}{1}$			
Hexagonal and Rhombohedral:	$hk \cdot l$	$hh \cdot l$	$0k \cdot l$	$hk \cdot 0$	$hh \cdot 0$	$0k \cdot 0$	$00 \cdot l$		
	$\frac{24^*}{1}$	$\frac{12^*}{8}$	$\frac{12^*}{8}$	$\frac{12^*}{8}$	$\frac{6}{6}$	$\frac{6}{6}$	$\frac{2}{2}$		
Tetragonal:	$hkl$	$hhl$	$0kl$	$hko$	$hko$	$0ko$	$00l$		
	$\frac{16^*}{8}$	$\frac{8}{8}$	$\frac{8}{8}$	$\frac{8^*}{8}$	$\frac{4}{4}$	$\frac{4}{4}$	$\frac{2}{2}$		
Orthorhombic:	$hkl$	$0kl$	$h0l$	$hko$	$h00$	$0ko$	$00l$		
	$\frac{8}{8}$	$\frac{4}{4}$	$\frac{4}{4}$	$\frac{4}{4}$	$\frac{2}{2}$	$\frac{2}{2}$	$\frac{2}{2}$		
Monoclinic:	$hkl$	$h0l$	$0ko$						
	$\frac{4}{4}$	$\frac{2}{2}$	$\frac{2}{2}$						
Triclinic:	$hkl$								
	$\frac{2}{2}$								

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# Peak Width



Scherrer formula

$$\text{FWHM} = \frac{0.88\lambda}{D \cos \theta} \quad (\text{radian})$$

Sources for broadening

Size

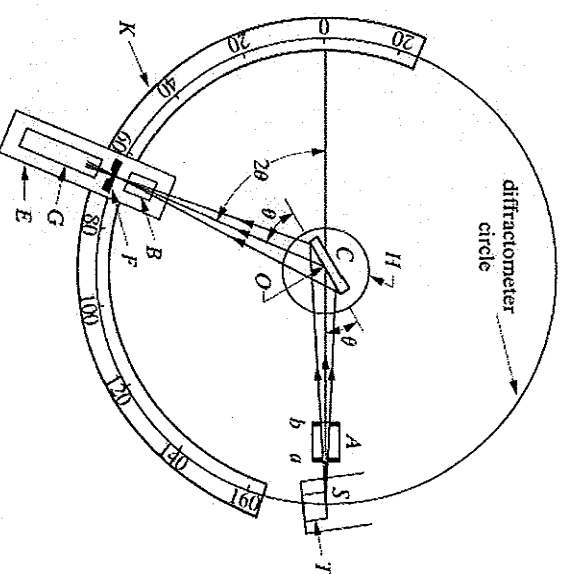
Instrument width

Stress

# Common XRD Techniques

Powder diffraction:

fixed  $\lambda$ , scan  $\theta$   
low angle reflectivity



Energy-dispersive:

fixed  $\theta$ , white radiation & high-reso. detectors

Laue method:

fixed  $\theta$ , white radiation

Debye Scherrer:

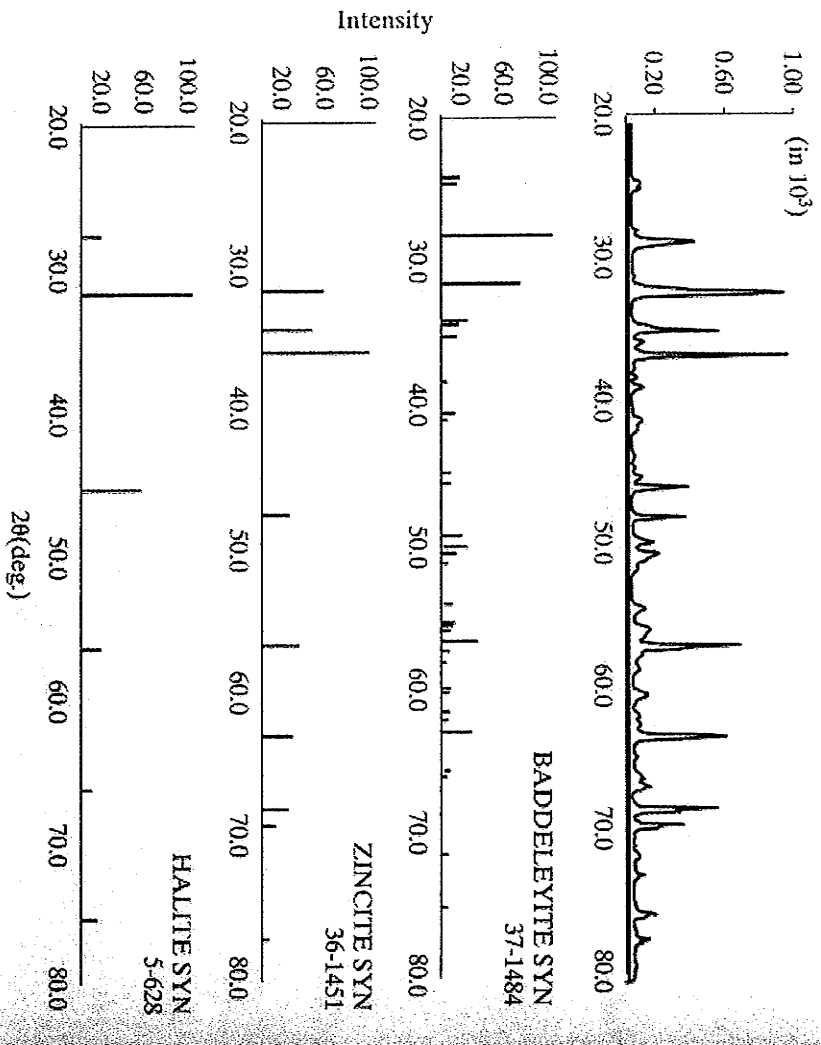
fixed  $\lambda$ , captures all  $\theta$ 's

Rotating crystal:

fixed  $\lambda$ , vary  $\theta$  mechanically



# Phase Identification



Account for all peaks:

position

relative intensity

peak width

